

# Synergistic Metal-Support Interactions in Au/GaN Catalysts for Photoelectrochemical Nitrate Reduction to Ammonia

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Metal-support interactions are crucial in the electrochemical synthesis of ammonia (NH<sub>3</sub>) from nitrate (NO<sub>3</sub><sup>-</sup>) reduction reaction, enabling efficient NH<sub>3</sub> production under mild conditions. However, the complexity of the reaction pathways often limits efficiency. Here, a photoelectrochemical system composed of gold (Au) nanoclusters supported on gallium nitride (GaN) nanowires is introduced, grown on a n<sup>+</sup>-p Si wafer, for selective reduction of NO<sub>3</sub><sup>-</sup> to NH<sub>3</sub> under solar illumination. NO<sub>3</sub><sup>-</sup> ions are preferentially adsorbed and reduced to nitrite (NO<sub>2</sub><sup>-</sup>) on the GaN nanowires, which then transfer to adjacent Au nanoclusters to complete the NH<sub>3</sub> synthesis. This mechanism is confirmed by both experimental data and theoretical calculations. Optimizing the surface coverage and size of Au nanoclusters on the GaN nanowires significantly enhanced catalytic activity compared to that on planar n<sup>+</sup>-p Si photoelectrodes, achieving a faradaic efficiency of 91.8% at -0.4 V<sub>RHE</sub> and a high NH<sub>3</sub> production rate of 131.1 μmol cm<sup>-2</sup> h<sup>-1</sup> at -0.8 V<sub>RHE</sub>. These findings highlight the synergetic effect between metal co-catalysts and semiconductor supports in designing photoelectrodes for multi-step NO<sub>3</sub><sup>-</sup> reduction.

## 1. Introduction

Ammonia (NH<sub>3</sub>) is a critical feedstock for fertilizers and a promising carbon-free hydrogen carrier, easily stored in liquid form.<sup>[1,2]</sup> Currently, industrial NH<sub>3</sub> production relies on the energy-intensive and carbon-emissive Haber–Bosch process, which operates under high temperatures and pressure.<sup>[3,4]</sup> As a clear alternative, the electrochemical nitrogen (N<sub>2</sub>) reduction reaction in aqueous electrolytes allows for NH<sub>3</sub> synthesis under milder conditions.<sup>[5–7]</sup> However, the high dissociation energy of the N≡N bond (941 kJ mol<sup>-1</sup>) limits both the selectivity and efficiency. In contrast, the electrochemical nitrate reduction reaction (NO<sub>3</sub><sup>-</sup>RR) is more feasible due to the significantly lower dissociation energy of the N=O bond (204 kJ mol<sup>-1</sup>) and the much higher solubility of NO<sub>3</sub><sup>-</sup> in aqueous electrolytes—~40000 times greater than that of N<sub>2</sub>.<sup>[8,9]</sup> Furthermore, the abundance of NO<sub>3</sub><sup>-</sup> in wastewater makes its reduction a sustainable approach to

NH<sub>3</sub> synthesis, while also providing an effective solution for mitigating environmental pollution through runoff purification.

The NO<sub>3</sub><sup>-</sup>RR to NH<sub>3</sub> is a complex eight-electron process (NO<sub>3</sub><sup>-</sup> + 9H<sup>+</sup> + 8e<sup>-</sup> → NH<sub>3</sub> + 3H<sub>2</sub>O) involving 9 protons (H<sup>+</sup>) for deoxygenation and hydrogenation steps.<sup>[10]</sup> Previous studies have explored various metal-based catalysts, including Cu-PTCDA,<sup>[11]</sup> strained Ru nanoclusters,<sup>[12]</sup> Fe single-atom catalysts,<sup>[13,14]</sup> and strained Bi,<sup>[15]</sup> to enhance the faradaic efficiency for NH<sub>3</sub> (FE<sub>NH3</sub>) and reduce the overpotential. While several catalysts have demonstrated success for NO<sub>3</sub><sup>-</sup>RR, detailed mechanistic studies remain limited.

Recently, binary metal compound catalysts such as CuNi,<sup>[16]</sup> CuCoSP,<sup>[17]</sup> Rh@Cu,<sup>[18]</sup> and NiCo<sub>2</sub>O<sub>4</sub><sup>[19]</sup> have shown improved NH<sub>3</sub> production compared over their single-metal counterparts, due to metal-support interactions and the transfer of reaction intermediates between multiple active phases. These heterogeneous catalysts enhance reaction kinetics and lower thermodynamic barriers by modulating the binding energy of adsorbates on the catalytic surface. For example, Rh clusters on Cu surfaces provide sufficient hydrogen, activating the hydrogenation process and promoting selective NH<sub>3</sub> production.<sup>[18]</sup> Similarly, Co(OH)<sub>2</sub> aids in the hydrogenation of adsorbed NO<sub>3</sub><sup>-</sup> in the CoP/Co(OH)<sub>2</sub> catalytic system.<sup>[20]</sup>

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In microorganisms,  $\text{NH}_3$  synthesis from  $\text{NO}_3^-$ -RR occurs through two tandem reactions: 1)  $\text{NO}_3^-$  reduction to  $\text{NO}_2^-$  by nitrate reductase, and 2)  $\text{NO}_2^-$  conversion to  $\text{NH}_3$  by nitrite reductase or nitrogenase.<sup>[21–23]</sup> This tandem process has been artificially replicated using CuCoSP, where Cu/CuO<sub>x</sub> phases reduce  $\text{NO}_3^-$  to  $\text{NO}_2^-$ , and the  $\text{NO}_2^-$  intermediate is transferred and converted to  $\text{NH}_3$  on Co/CoO phases.<sup>[17]</sup> The design strategy of combining dual active sites for tandem reactions can also be applied to metal-semiconductor systems. Successfully designing such systems could leverage photon-to-electron conversion in semiconductors to further reduce the overpotential required for  $\text{NO}_3^-$ -RR, as demonstrated in previous energy-efficient photoelectrochemical reactions.<sup>[24–31]</sup>

In this study, we investigated the synergistic metal-support interactions between Au nanoclusters (NCs) and GaN nanowires (NWs) grown on an  $n^+$ -p Si wafer for photoelectrochemical (PEC)  $\text{NO}_3^-$ -RR to  $\text{NH}_3$  under simulated solar light. By optimizing the surface coverage and size of Au NCs on GaN NWs, we achieved rapid adsorption and reduction of  $\text{NO}_3^-$  to  $\text{NO}_2^-$  on the GaN NWs, followed by efficient  $\text{NH}_3$  generation on the Au NCs. This optimization led to a significant increase in the Faradaic efficiency for  $\text{NH}_3$  ( $\text{FE}_{\text{NH}_3}$ ) reaching 91.8% at  $-0.4$  V vs reversible hydrogen electrode ( $V_{\text{RHE}}$ ). Density functional theory (DFT) calculations corroborated the experimental findings, showing that  $\text{NO}_3^-$  is reduced to  $\text{NO}_2^-$  on the GaN surface and displaced by the remaining  $\text{NO}_3^-$ , while  $\text{NO}_2^-$  is further reduced to  $\text{NH}_3$  on the Au NCs. The sequential reduction reactions occurring on the adjacent Au and GaN surfaces resulted in a high production rate of  $\text{NH}_3$  ( $Y_{\text{NH}_3}$ ) of  $131.1 \mu\text{mol cm}^{-2} \text{h}^{-1}$  at  $-0.8 V_{\text{RHE}}$  with selective  $\text{NH}_3$  generation ( $\text{FE}_{\text{NH}_3} > 90\%$ ) sustained for 8 h at  $-0.4 V_{\text{RHE}}$ . This work offers valuable insights for designing metal/semiconductor systems to achieve high-efficiency PEC  $\text{NO}_3^-$ -RR to  $\text{NH}_3$  under solar illumination.

## 2. Results And Discussion

Au/GaN/Si photoelectrodes were fabricated using plasma-assisted molecular beam epitaxy to grow GaN NWs on an  $n^+$ -p Si wafer, followed by photodeposition of Au NCs as described in the experimental section. Scanning electron microscopy (SEM) studies confirmed the vertical growth of GaN NWs on the planar  $n^+$ -p Si wafer (Figure 1a). The GaN NWs measured  $\approx 450$  nm in length and 50 nm in diameter. Three variations of Au/GaN/Si photoelectrodes were prepared by adjusting the volume of 0.4 M  $\text{HAuCl}_4$  precursor solution used during photodeposition, denoted as Au#/GaN/Si, where # indicates the precursor solution volume. As the precursor volume increased from 1 to 100  $\mu\text{L}$ , the size of Au NCs increased, particularly in the upper region of the GaN NWs (Figure S1, Supporting Information). The microstructure of GaN NWs and Au NCs was further examined using a high-angle annular dark-field scanning transmission electron microscopy (HAADF-STEM) (Figure 1b; Figure S2, Supporting Information). Due to the significant atomic number difference between Au (79) and Ga (31), Z-contrast imaging revealed locally aggregated Au NCs on both the top and sidewalls of the GaN NWs. Elemental mapping by energy-dispersive X-ray spectroscopy (EDS) showed a uniform distribution of Ga (blue), and N (green) across the GaN NWs, while Au NCs (pink), ranging from 2 to 20 nm in size, were sparsely distributed on the GaN sur-

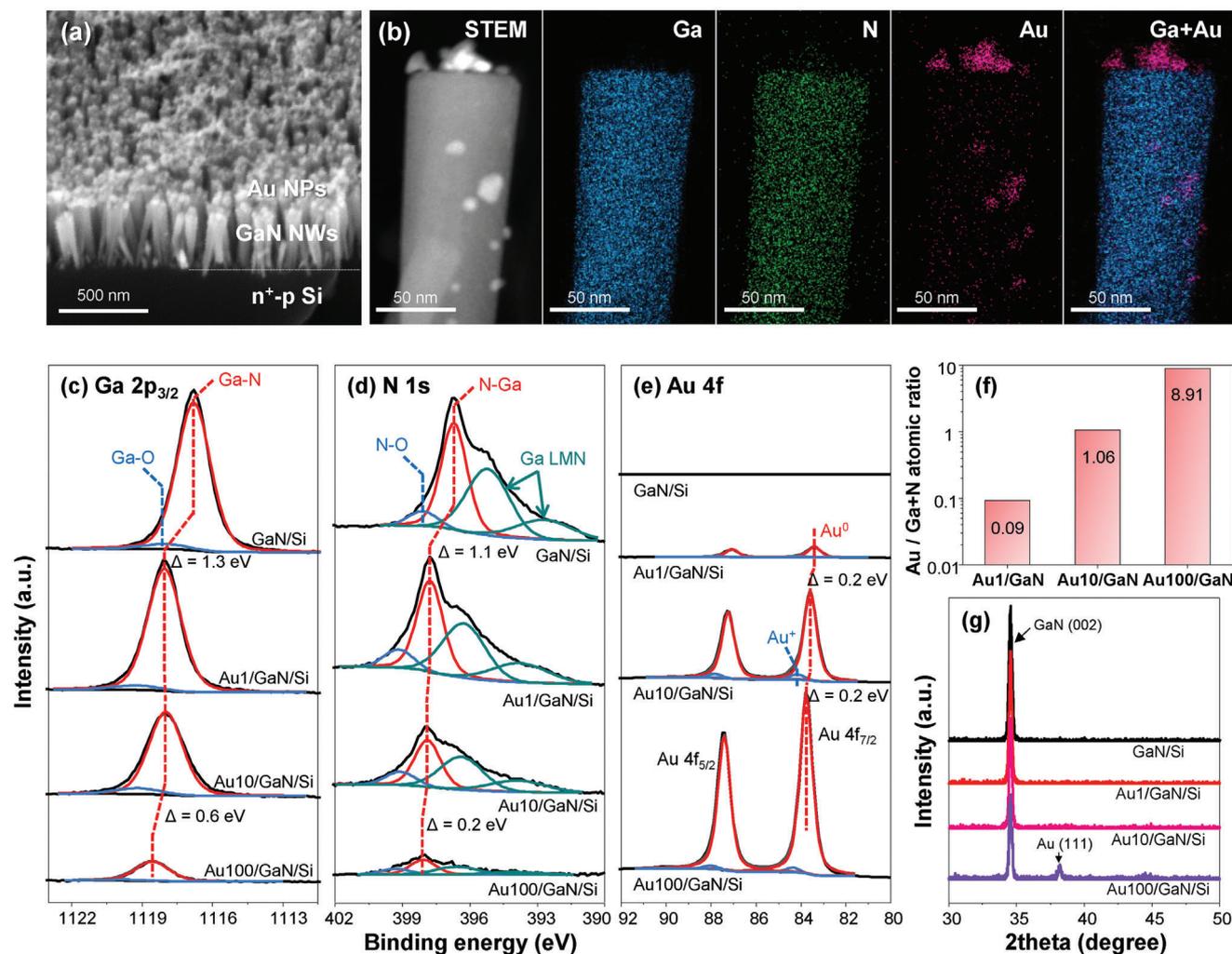
face, indicating a non-uniform decoration of Au NCs on the GaN NWs. The non-uniform distribution of Au NCs can be attributed to preferential photodeposition at the tips of GaN NWs, where enhanced light absorption and higher photo-excited electron density promote localized growth. In future work, fine-tuning the photodeposition parameters, such as light wavelength and deposition time, will be pursued to achieve the formation of more uniformly distributed Au NC.

X-ray photoelectron spectroscopy (XPS) was used to analyze the surface chemical composition of the photoelectrodes. The Ga  $2p_{3/2}$  XPS spectrum was deconvoluted into a dominant Ga–N bond peak at 1116.8 eV and a minor Ga–O bond peak at 1118.0 eV (Figure 1c).<sup>[32]</sup> As the amount of Au NCs increased, the intensity of the Ga  $2p_{3/2}$  signal diminished due to the screening effect of the Au NCs, which blocked photoelectrons emitted from the GaN surface. Similarly, the N 1s XPS spectrum (Figure 1d) was split into a major N–Ga bond at 396.7 eV and a minor N–O bond at 398.1 eV,<sup>[33]</sup> both of which showed decreasing intensity with increasing Au NC loading. The Au NCs exhibited the metallic Au<sup>0</sup> phase, with binding energies at 83.4 and 87.1 eV, alongside a trace amount of Au<sup>+</sup> at 84 and 87.7 eV (Figure 1e). As the Au loading increased, the intensity of the Au 4f peaks increased and the Au 4f peak shifted toward a higher binding energy, approaching the reference value of 84 eV for bulk Au  $4f_{7/2}^0$ .<sup>[34]</sup> This indicates that smaller Au NCs have weaker electron binding compared to bulk Au, primarily due to electron transfer from GaN NWs to Au NCs at the interface. This interfacial charge transfer is particularly pronounced for smaller Au NCs, as shown by the shifts of Ga and N peaks to higher binding energies and Au peaks to lower binding energies in the XPS spectra. As the size of the Au NCs increases, thicker Au layers gradually screen the interfacial charge transfer effect, leading to bulk-like electronic properties.

The surface atomic ratios of Au with respect to Ga and N, calculated from the XPS data, were 0.09, 1.06, and 8.91 for Au1/GaN, Au10/GaN, and Au100/GaN, respectively (Figure 1f).

X-ray diffraction (XRD) pattern of GaN/Si revealed a distinct GaN (002) peak [JCPDS #02-1078], confirming GaN NWs were single crystals (Figure 1g). The Au (111) peak was only detected in the Au100/GaN/Si sample, indicating that the crystallite size of the Au NCs was sufficiently large for XRD detection [JCPDS #04-0784]. In contrast, smaller Au NCs such as Au1 and Au10 samples, likely had a higher proportion of surface-exposed Au atoms, particularly at terrace sites, compared to bulk crystal facets. This increased surface exposure, especially on high-index crystal planes, can enhance catalytic properties by providing a greater number of active sites, potentially enhancing the reactivity.

PEC  $\text{NO}_3^-$ -RR was performed in an H-type cell using simulated AM1.5G 1-sun solar light, with the light sources positioned perpendicular to the planar  $n^+$ -p Si photoelectrodes. Linear sweep voltammetry (LSV) curves were recorded in 0.1 M  $\text{K}_2\text{SO}_4$  with varying  $\text{KNO}_3$  concentrations (0, 0.02, 0.1, 0.5, and 1 M). As the  $\text{NO}_3^-$  concentration increased from 0 to 1 M, the onset potential ( $V_{\text{onset}}$ ) at  $-1.0 \text{ mA cm}^{-2}$  for GaN/Si shifted significantly from  $-0.63$  to  $-0.21 V_{\text{RHE}}$  (Figure 2a), indicating that GaN/Si exhibits higher catalytic activity for  $\text{NO}_3^-$ -RR compared to the hydrogen evolution reaction (HER). However, as the loading of Au NCs increased, the difference in  $V_{\text{onset}}$  between electrolytes with different  $\text{NO}_3^-$  concentrations diminished (Figure S3, Supporting

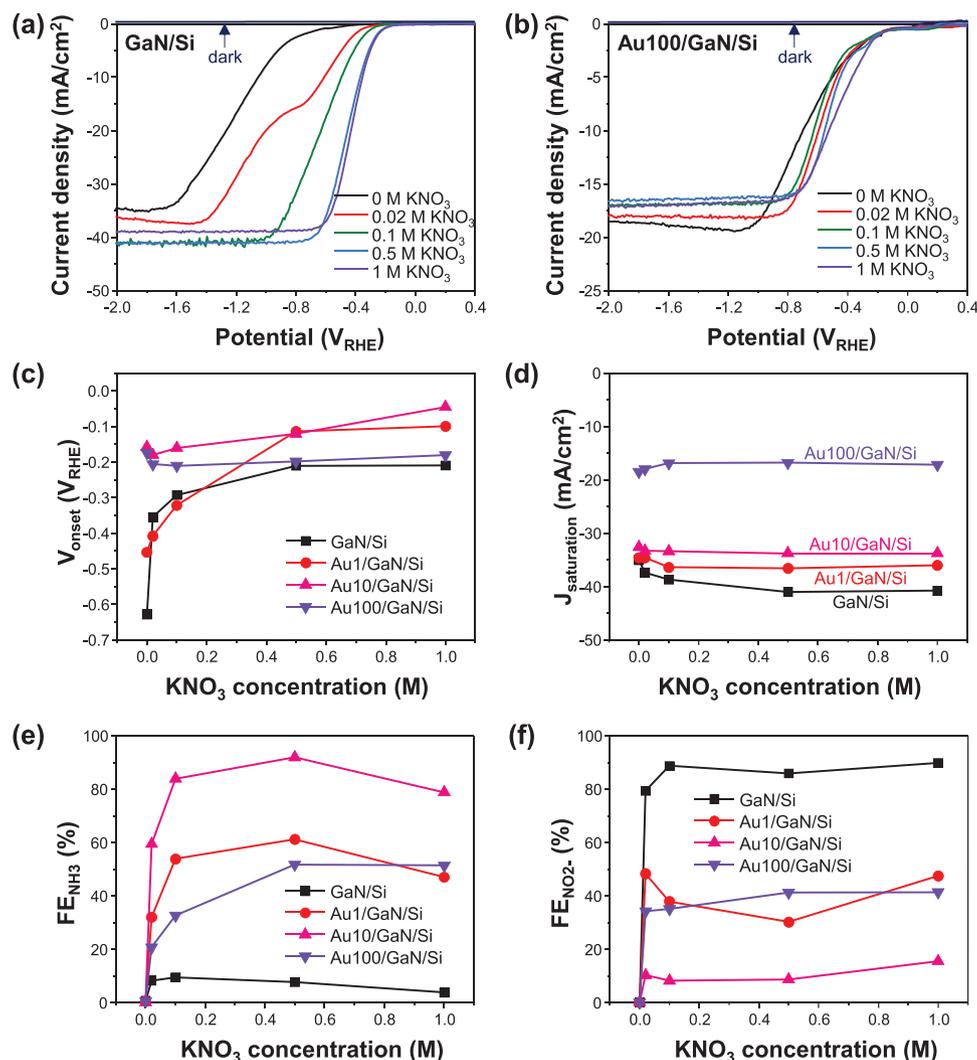


**Figure 1.** a) 45°-tilt-view SEM image of Au10/GaN/Si. b) STEM image and EDS elemental maps of Au/GaN. Ga (blue) and N (green) from GaN NWs were uniformly distributed while Au (pink) from Au NCs was aggregated on the upper region of GaN NWs. XPS spectra of c) Ga 2p<sub>3/2</sub>, d) N 1s, e) Au 4f for GaN/Si, Au1/GaN/Si, Au10/GaN/Si, and Au100/GaN/Si. f) Au/Ga+N ratio calculated from XPS analysis. g) XRD patterns of photoelectrodes.

Information). Eventually, Au100/GaN/Si showed minimal variation in LSV curves regardless of NO<sub>3</sub><sup>-</sup> concentration, with V<sub>onset</sub> ranging between -0.17 to -0.21 V<sub>RHE</sub> (Figure 2b,c). These results suggest competitive behavior between NO<sub>3</sub><sup>-</sup>RR and HER on the Au 100 co-catalyst. Additionally, the saturated photocurrent density (J<sub>saturation</sub>) at -2 V<sub>RHE</sub> decreased from -41.0 to -16.8 mA cm<sup>-2</sup> as the Au NCs loading increased, likely due to the Au NCs blocking incident light and reducing light absorption by the underlying n<sup>+</sup>-p Si photoelectrode (Figure 2d).<sup>[35]</sup> Under dark conditions, all photoelectrodes exhibited a current density close to zero, attributed to the absence of photon-induced charge carrier generation. This indicates that the reduction reactions on the photoelectrodes are entirely reliant on photo-generated charge carriers. To further investigate photovoltage generation by photoelectrode, we fabricated an electrochemical reaction electrode by growing GaN NWs on an electrically conductive n-type silicon substrate (n-Si) and loading it with Au NCs. LSV curves measured in 0.1 M K<sub>2</sub>SO<sub>4</sub> with 0.1 M KNO<sub>3</sub> electrolyte revealed that the onset potential for the Au/GaN/n-Si electrode under dark conditions was

≈0.5 V more negative compared to the Au/GaN/n<sup>+</sup>-p Si photoelectrode under illuminated conditions (Figure S4, Supporting Information). This difference highlights the role of underlying n<sup>+</sup>-p Si in reducing the voltage requirement by effectively harnessing solar energy.

Faradaic efficiencies (FEs) for GaN/Si, Au1/GaN/Si, Au10/GaN/Si, and Au100/GaN/Si were assessed in electrolytes with varying NO<sub>3</sub><sup>-</sup> concentrations at -0.4 V<sub>RHE</sub> (Figure 2e,f). Concentrations of NH<sub>3</sub>, NO<sub>2</sub><sup>-</sup>, and N<sub>2</sub>H<sub>4</sub> products were determined via spectrophotometric analysis (Figure S5, Supporting Information). In 0.1 M K<sub>2</sub>SO<sub>4</sub> (without NO<sub>3</sub><sup>-</sup>), all photoelectrodes exhibited selective HER with FE<sub>H<sub>2</sub></sub> > 95% (Figure S6, Supporting Information). Upon introducing 0.02 M KNO<sub>3</sub>, the production of NH<sub>3</sub>, NO<sub>2</sub><sup>-</sup>, and N<sub>2</sub>H<sub>4</sub> was observed, with Au10/GaN/Si achieving the highest FE<sub>NH<sub>3</sub></sub> of 59.5% (Figure 2e). Notably, GaN/Si effectively suppressed HER (FE<sub>H<sub>2</sub></sub> < 3%) and selectively converted NO<sub>3</sub><sup>-</sup> into NO<sub>2</sub><sup>-</sup> (FE<sub>NO<sub>2</sub><sup>-</sup></sub> > 80%), even with 0.02 M NO<sub>3</sub><sup>-</sup> (Figure 2f; Figure S7, Supporting Information). However, H<sub>2</sub> production remained significant on Au/GaN/Si,



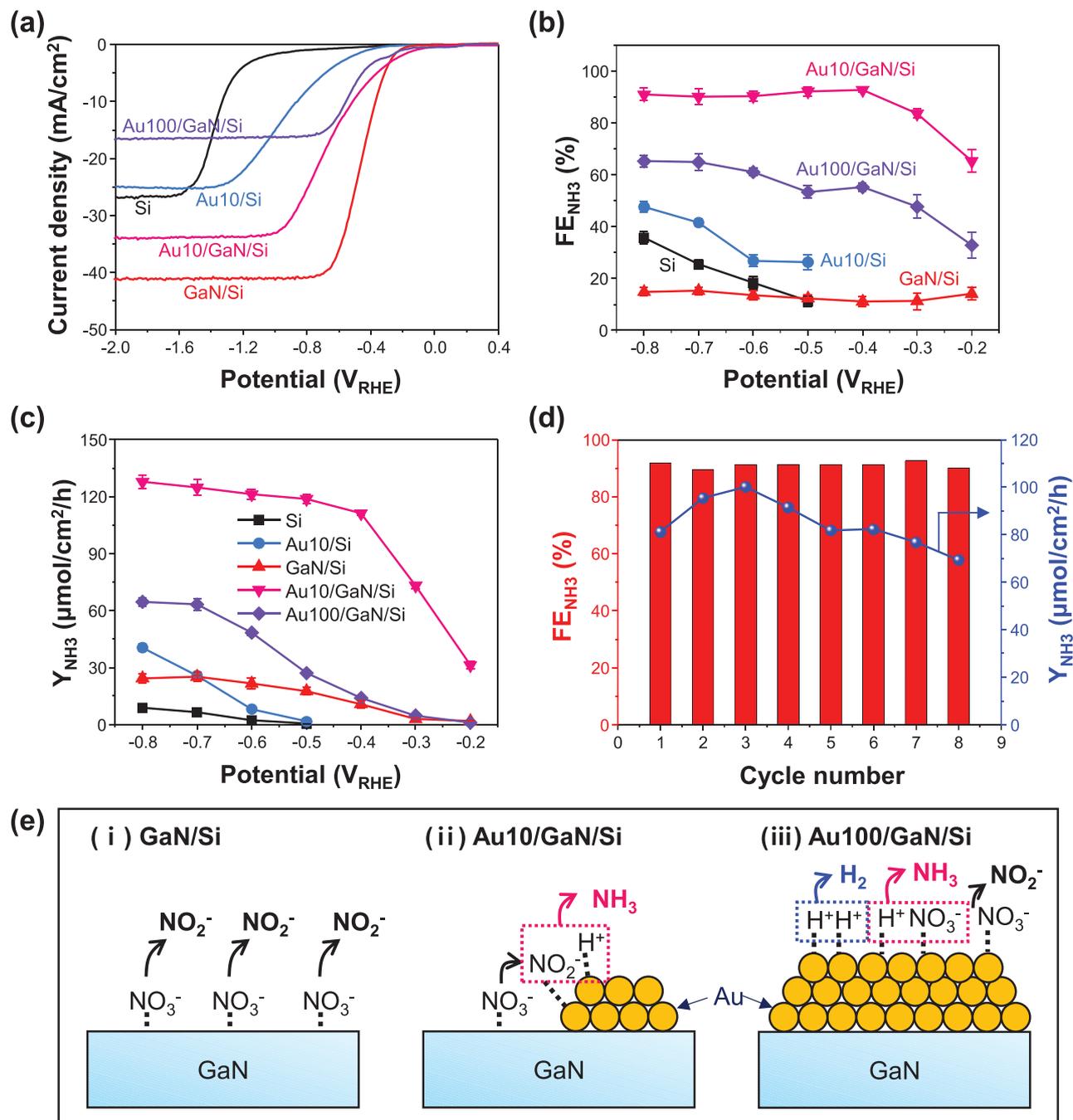
**Figure 2.** LSV curves of a) GaN/Si and b) Au100/GaN/Si measured in 0.1 m  $K_2SO_4$  with  $KNO_3$  concentrations of 0, 0.02, 0.1, 0.5, and 1 m. The measurement was conducted under 1-sun-light illumination. Under dark conditions, the photoelectrodes were tested in 0.1 m  $K_2SO_4$  electrolyte. Plots of c)  $V_{onset}$  at  $-1.0 \text{ mA cm}^{-2}$  and d)  $J_{saturation}$  at  $-2 V_{RHE}$  with  $KNO_3$  concentration. Faradaic efficiency of e)  $NH_3$  and f)  $NO_2^-$  for GaN/Si, Au1/GaN/Si, Au10/GaN/Si, and Au100/GaN/Si measured in 0.1 m  $K_2SO_4$  with different  $KNO_3$  concentration at  $-0.4 V_{RHE}$ . Au10/GaN/Si represented the highest  $FE_{NH_3}$  (91.8%) among the tested samples.

with the highest  $FE_{H_2}$  of 41.6% observed for Au100/GaN/Si (Figure S6, Supporting Information). As  $NO_3^-$  concentration increased, the  $FE_{NH_3}$  for Au/GaN/Si gradually rose (Figures S8–S10, Supporting Information). Au10/GaN/Si demonstrated the optimal balance of Au coverage, achieving a maximum  $FE_{NH_3}$  of 91.8% at 0.5 m  $NO_3^-$ , highlighting the importance of fine-tuning Au/GaN metal-support interactions to maximize catalytic performance.

To further investigate metal-support interactions, we fabricated planar  $n^+p$  Si and Au-coated  $n^+p$  Si (Au10/Si) photoelectrodes. The LSV curve of Au10/Si exhibited a more positive  $V_{onset} = -0.41 V_{RHE}$  compared to Si ( $-0.81 V_{RHE}$ ) in 0.1 m  $K_2SO_4$  with 0.5 m  $KNO_3$  (Figure 3a). However, both remained more negative than those of GaN/Si or Au/GaN/Si, highlighting GaN NWs as promising platforms for loading Au co-catalysts in  $NO_3^-$ RR. Ad-

ditionally, the  $J_{saturation}$  of GaN/Si was  $\approx 1.5$  times higher than planar Si, attributed to the antireflective properties of GaN NWs.<sup>[36]</sup>

The influence of the applied potential on FE and production rates was evaluated for Si (Figure S11, Supporting Information), Au/Si (Figure S12, Supporting Information), GaN/Si (Figure S13, Supporting Information), Au10/GaN/Si (Figure S14, Supporting Information), and Au100/GaN/Si (Figure S15, Supporting Information), with a summary of  $FE_{NH_3}$  shown in Figure 3b. Planar Si achieved a  $FE_{NH_3}$  of 36.3% at  $-0.8 V_{RHE}$ , while Au deposition on Si (Au/Si) marginally increased the  $NH_3$  selectivity ( $FE_{NH_3} = 47.2\%$  at  $-0.8 V_{RHE}$ ). In contrast, GaN/Si predominantly produced  $NO_2^-$  ( $FE_{NO_2^-} > 90\%$  at potential  $< -0.3 V_{RHE}$ ), with low  $NH_3$  ( $FE_{NH_3} < 16\%$ ) and  $N_2H_4$  ( $FE_{N_2H_4} < 3\%$ ) selectivities. GaN NWs effectively reduce  $NO_3^-$  to  $NO_2^-$ , which was desorbed from the photoelectrode surface before further



**Figure 3.** LSV curves of Si, Au10/Si, GaN/Si, Au10/GaN/Si, and Au100/GaN/Si measured in 0.1 M K<sub>2</sub>SO<sub>4</sub> with 0.5 M KNO<sub>3</sub>. b) Faradaic efficiency and c) production rate of NH<sub>3</sub> plotted as a function of cathodic potentials from -0.2 to -0.8 V<sub>RHE</sub>. NH<sub>3</sub> production was measured three times, and the mean values along with their standard deviations are represented as error bars. Au10/GaN/Si had the widest potential window (-0.4 to -0.8 V<sub>RHE</sub>) for high FE<sub>NH<sub>3</sub></sub> > 90% and achieved the best Y<sub>NH<sub>3</sub></sub> = 131.1 μmol cm<sup>-2</sup> h<sup>-1</sup> at -0.8 V<sub>RHE</sub>. d) FE<sub>NH<sub>3</sub></sub> and Y<sub>NH<sub>3</sub></sub> for 8 cycles of reaction. Each cycle was performed for 1 h at -0.4 V<sub>RHE</sub>. e) Schematic illustrations of adsorption of NO<sub>3</sub><sup>-</sup> and NO<sub>3</sub><sup>-</sup> reduction reaction on (i) GaN/Si, (ii) Au10/GaN/Si, and (iii) Au100/GaN/Si. The dashed line represents the adsorption of molecules and the arrow line indicates the reduction reaction.

reduction. Au10/GaN/Si demonstrated a broad potential window (-0.4 to -0.8 V<sub>RHE</sub>) for selective NH<sub>3</sub> production (FE<sub>NH<sub>3</sub></sub> > 90%). As a result, Au10/GaN/Si exhibited the highest FE<sub>NH<sub>3</sub></sub> among the tested photoelectrodes, achieving NH<sub>3</sub> production rates (Y<sub>NH<sub>3</sub></sub>) of 110 μmol cm<sup>-2</sup> h<sup>-1</sup> at -0.4 V<sub>RHE</sub> and 131.1 μmol cm<sup>-2</sup> h<sup>-1</sup> at -0.8 V<sub>RHE</sub> (Figure 3c). The FE<sub>NH<sub>3</sub></sub> and Y<sub>NH<sub>3</sub></sub> values for Au10/GaN/Si

outperformed recent photoelectrodes and were comparable to state-of-the-art electrocatalysts (Table S1, Supporting Information).

We assessed the stability of Au10/GaN/Si over 8 cycles in 0.1 M K<sub>2</sub>SO<sub>4</sub> with 0.5 M KNO<sub>3</sub> (Figures 3d). Each cycle was run for 1 h at a constant potential of -0.4 V<sub>RHE</sub>, with fresh electrolytes

introduced for every cycle. Chronoamperometric curve (Figure S16a, Supporting Information) revealed an initial high photocurrent density, which gradually decreased over 1 h of reaction. This behavior was likely due to the depletion of  $\text{NO}_3^-$  concentration near the surface as the reaction progresses, and the accumulation of by-products that poisoned the surface. Upon replacing the electrolyte with a fresh solution, the photocurrent density was restored to its initial high level, confirming that the observed decrease was primarily due to changes in the reaction environment rather than irreversible degradation of the photoelectrode. Throughout the 8 cycles, a high  $\text{NH}_3$  faradaic efficiency ( $\text{FE}_{\text{NH}_3} > 90\%$ ) was consistently maintained (Figure S16b, Supporting Information). The  $\text{NH}_3$  production rate ( $Y_{\text{NH}_3}$ ) increased to  $100.1 \mu\text{mol cm}^{-2} \text{h}^{-1}$  during the first 3 cycles but gradually declined to  $69.3 \mu\text{mol cm}^{-2} \text{h}^{-1}$  over the last 5 cycles (Figure S16c, Supporting Information). This reduction in  $Y_{\text{NH}_3}$  was attributed to the detachment of Au NCs from the GaN NWs (Figure S17, Supporting Information). Although Au NCs remained on the surface of GaN NWs after 8 cycles, the Au/Ga+N ratio dropped from 1.06 to 0.47, indicating a loss of Au NCs. To address this issue, incorporating protective layers,<sup>[37,38]</sup> or applying pre- or post-treatments<sup>[30]</sup> could be explored to enhance the interfacial binding between Au NCs and GaN NWs, thereby improving the long-term stability of the photoelectrode. Despite this challenge, Au10/GaN/Si remained a promising photoelectrode for selective and sustained  $\text{NH}_3$  production with extended stability. Additionally, isotope labeling experiments confirmed that  $\text{NH}_3$  originated from  $\text{NO}_3^-$  rather than from the nitrogen of GaN, demonstrating that GaN NWs remained chemically stable during the PEC  $\text{NO}_3^-$  RR (Figure S18, Supporting Information).

Our findings highlight that initial  $\text{NO}_3^-$  adsorption is critical for selective  $\text{NH}_3$  production. On the GaN surface,  $\text{NO}_3^-$  primarily adsorbs and is converted to  $\text{NO}_2^-$ , which desorbs without further reduction, leading to a high  $\text{FE}_{\text{NO}_2^-} > 80\%$  (Figure 3e). However, when a moderate amount of small Au NCs was deposited on the GaN surface (Au10/GaN/Si), the  $\text{NO}_2^-$  produced on GaN migrates to nearby Au sites where further reduction to  $\text{NH}_3$  occurs via deoxygenation and hydrogenation with  $\text{H}^+$ , resulting in high  $\text{FE}_{\text{NH}_3} > 90\%$  (Figure S19a, Supporting Information). In contrast, larger Au crystallites with higher surface coverage on GaN (Au100/GaN/Si) impaired  $\text{NO}_3^-$  adsorption (Figure S19b, Supporting Information) and reduced the number of Au-GaN active sites. This led to competition between  $\text{NO}_3^-$  RR and the hydrogen evolution reaction (HER), causing a drop in  $\text{FE}_{\text{NH}_3}$  to below 70%.

The mechanistic insights from DFT calculations support these observations (see free energies of reactants, products, and intermediates in Tables S2 and S3, Supporting Information). Previous research on photoelectrochemical  $\text{NO}_3^-$  reduction on GaN nanowires has demonstrated that  $\text{NO}_2^-$  is the primary product.<sup>[39]</sup> This is because the reduction of  $\text{NO}_3^-$  to  $\text{NO}_2^-$  is thermodynamically favorable, requiring only one endergonic electron transfer step with an energy of +0.29 eV (as reported in ref. [39]), which is easily provided by the photo-excited electrons. In contrast, the reduction of  $\text{NO}_2^-$  to NO to form the first intermediate toward reduction to  $\text{NH}_3$  requires significantly more energy (+0.65 eV) (Figure 4a). Additionally, GaN shows a stronger affinity for  $\text{NO}_3^-$  over  $\text{NO}_2^-$ , with the binding free energy of  $\text{NO}_3^-$  being more favorable by  $\approx -0.09$  eV. This preferential binding of  $\text{NO}_3^-$  leads to the displacement of  $\text{NO}_2^-$ , which explains why

$\text{NO}_2^-$  is the main product of  $\text{NO}_3^-$  reduction. Thus, the combination of lower energy requirements for  $\text{NO}_3^-$  reduction and the stronger binding affinity of  $\text{NO}_3^-$  over  $\text{NO}_2^-$  results in  $\text{NO}_2^-$  being expelled as the main product during  $\text{NO}_3^-$  reduction.

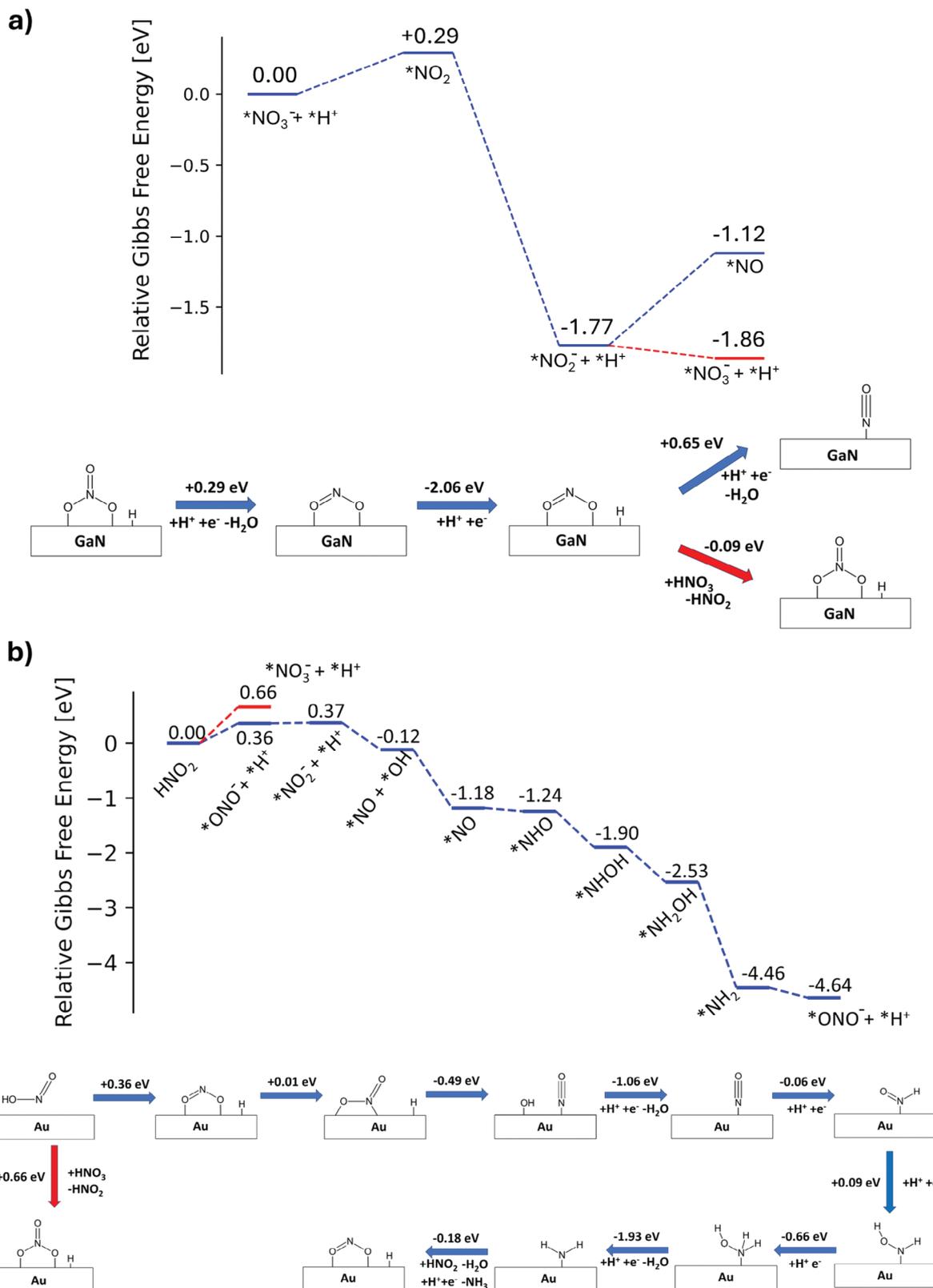
On the Au surface (Figure S20, Supporting Information), the binding preference between  $\text{NO}_3^-$  and  $\text{NO}_2^-$  is reversed: we find that  $\text{NO}_2^-$  exhibits a more negative binding free energy than  $\text{NO}_3^-$  by  $\approx -0.30$  eV on the Au (211) surface (Figure S21, Supporting Information). This suggests that once  $\text{NO}_2^-$  has been formed, it preferentially binds to Au making the initial adsorption of  $\text{NO}_3^-$  on Au (211) more difficult. Additionally, the reduction of  $\text{NO}_3^-$  to  $\text{NO}_2^-$  on Au involves an endergonic step of +0.30 eV during the conversion of  $^*\text{NO}_2$  to  $^*\text{NO}_2^-$  (Figure S22, Supporting Information), indicating that  $\text{NO}_3^-$  reduction to  $\text{NO}_2^-$  on Au is relatively slow and easily outcompeted by GaN. It is also worth noting that while both  $\text{NO}_2^-$  and  $\text{NO}_3^-$  chemisorption on the inert Au surface is entropically disfavored in a vacuum, the chemisorption of  $\text{NO}_2^-$  is more favorable than that of  $\text{NO}_3^-$ .

In contrast, the further reduction of  $\text{NO}_2^-$  is strongly favored on Au (Figure 4b). This process involves a transition in the binding mode from  $^*\text{ONO}^-$  (where  $\text{NO}_2^-$  binds via its two oxygen atoms to the Au step) to the  $^*\text{NO}_2^-$  (where nitrogen binds to the Au step, with one of the oxygen atoms pointing down and adsorbing onto the lower terrace) (Figure S23, Supporting Information). This switch from oxygen to nitrogen binding is crucial for the dissociation of the second N=O bond and has been extensively investigated in other metal systems.<sup>[40]</sup> Here, a minimal free energy of 0.01 eV –easily surmountable thermally– enables irreversible bond dissociation to  $^*\text{NO}$  and  $^*\text{OH}$  intermediates on the surface. The subsequent removal of  $^*\text{OH}$  through protonation, coupled with the reduction of the oxidized Au surface, is highly exergonic.

Further reduction steps to  $^*\text{NHO}$ ,  $^*\text{NHOH}$ ,  $^*\text{NH}$ , and  $^*\text{NH}_2$ , culminating in the release of  $\text{NH}_3$  are also exergonic, leading to complete reduction to  $\text{NH}_3$  once  $\text{NO}_2^-$  dissociates. These computational findings align well with experimental observations, explaining why Au NCs without GaN show low  $\text{NH}_3$  production yield—since little  $\text{NO}_2^-$  is formed and made available for further reduction – and why GaN alone exhibits high faradaic efficiency toward  $\text{NO}_2^-$  rather than to  $\text{NH}_3$ . High and selective production of  $\text{NH}_3$  is observed only when GaN and Au are used together, with GaN reducing  $\text{NO}_3^-$  to  $\text{NO}_2^-$ , which desorbs and undergoes further reduction on Au, ultimately forming  $\text{NH}_3$ .

### 3. Conclusions

In conclusion, we have demonstrated the unique metal-support interactions between Au NCs and GaN NWs, which act as synergistic catalytic sites for PEC  $\text{NO}_3^-$  RR to  $\text{NH}_3$ . In this binary system,  $\text{NO}_3^-$  is preferentially reduced to  $\text{NO}_2^-$  on GaN NWs, while  $\text{NO}_2^-$  is subsequently converted to  $\text{NH}_3$  on adjacent Au NCs. The sequential reduction of  $\text{NO}_3^-$  and  $\text{NH}_3$  synthesis occurs across two adjacent Au/GaN phases, making the optimization of Au NCs coverage on GaN NWs crucial for achieving selective  $\text{NH}_3$  production. As a result, the designed Au/GaN/Si photoelectrode exhibited excellent performance, with faraday efficiency ( $\text{FE}_{\text{NH}_3}$ ) of 91.8% at  $-0.4 V_{\text{RHE}}$  and a high production rate ( $Y_{\text{NH}_3}$ ) of  $131.1 \mu\text{mol cm}^{-2} \text{h}^{-1}$  at  $-0.8 V_{\text{RHE}}$ , with no significant degradation over 8 h. A deeper understanding and precise design of the electronic and catalytic properties at the metal-semiconductor



**Figure 4.** a) Relative free energies of  $\text{NO}_3^-$  reduction to  $\text{NO}_2^-$  on GaN surfaces, with values taken from ref. [39]. After reduction to  $\text{NO}_2^-$ , the reduction to NO (and then further to  $\text{NH}_3$ ) shown in blue requires high reductive potentials, while expulsion of the  $\text{NO}_2^-$  by incoming  $\text{NO}_3^-$  is favorable (red), making  $\text{NO}_2^-$  the main product in PEC  $\text{NO}_3^-$  reduction on GaN. b) Relative free energies of  $\text{NO}_2^-$  reduction on Au (211) surface. Expulsion of the  $\text{NO}_2^-$  by  $\text{NO}_3^-$  (red) is unfavorable, meaning  $\text{NO}_2^-$  can be fully reduced toward  $\text{NH}_3$ .

interface will likely enable more efficient solar-driven  $\text{NO}_3^-$  RR to  $\text{NH}_3$ .

## Supporting Information

Supporting Information is available from the Wiley Online Library or from the author.

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W.J.D. and J.P.M. contributed equally to this work. This work was supported by the United States Army Research Office Award W911NF2110337.

## Conflict of Interest

Some IP related to this work was licensed to NS Nanotech, Inc. and NX Fuels, Inc., which were co-founded by Z. Mi. The University of Michigan and MI have a financial interest in the company.

## Data Availability Statement

The data that support the findings of this study are available from the corresponding author upon reasonable request.

## Keywords

ammonia, Au, GaN, metal-support interaction, nitrate reduction

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